Preparation of $SnO₂$ nanorods by annealing $SnO₂$ powder in NaCl flux

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 $SnO₂$ nanorods having the rutile structure have been prepared by annealing fine $SnO₂$ powder in a NaCl flux. The starting SnO₂ powder, the NaCl flux, and surfactant NP9 were mixed and heated at 800 °C for 2.5 h. The nanorods have diameters of ca. 20–40 nm and lengths of up to 1 μ m. High-resolution transmission electron microscopy (HRTEM), selected area electron diffraction (SAED) and X-ray diffraction (XRD) showed that the nanorods were well-crystallized with a rutile structure. The structure features and chemical composition of the as-prepared nanorods were analyzed by XRD, TEM, HRTEM, SAED, EDS and FTIR. A possible growth mechanism of the nanorods was described by the studies of the formation of nanorods with comparative experiments. The effects of NaCl and NP9 are discussed in detail.

1 Introduction

One-dimensional (1D) nanoscale materials have stimulated great interest recently both because of their unique electronic, optical, and mechanical properties and because of their potential applications in nanodevices.1,2 Many attempts have been made to synthesize one-dimensional nanostructure materials using a variety of nanofabrication techniques and crystal growth methods, such as arc discharge,³ laser ablation,⁴ template,^{5,6} solution,^{7–9} and other methods.^{10–12} However, to our knowledge, higher temperatures, special conditions, tedious procedures, long synthesis times, complex apparatus or the use of noxious gas compounds may be required for these methods. It has been and still is a challenge to find a novel and simple synthetic route for one-dimensional nano-scale materials.

Semiconductor one-dimensional nano-sized materials are known to have many interesting physical properties and great applications in optoelectronic devices, solar energy conversion, nonlinear optical, photoelectrochemical cells and heterogeneous photocatalysis.^{13,14} The rutile form of $SnO₂$ is one of the important materials. It is an n-type semiconductor with a large band gap, and is well-known for its applications in gas sensors,^{15,16} dye-based solar cells,¹⁷ transistors,¹⁸ electrode materials,¹⁹ catalysts,²⁰ and optical conductive films for solar cells,²¹ In recent years, attention has been focused on its possible applications in electrochromic devices.²²

 $SnO₂$ has been synthesized by many methods, such as sol–gel, 23,24 chemical vapor deposition, 25 magnetron sputtering,²⁶ and evaporation of elemental tin in an oxygen atmosphere,²⁷etc. However, only $SnO₂$ particles or films were prepared by these methods.

In this paper, we report a simple route for the preparation of $SnO₂$ nanorods. $SnO₂$ nanorods are synthesized by annealing precursor $SnO₂$ powder in the NaCl flux. The as-prepared nanorods are $ca. 20-40$ nm in diameter and up to 1 μ m in length. The nanorods are characterized by XRD, TEM, HRTEM, SAED, EDS, and FTIR. The growth mechanism

of SnO2 nanorods and the effects of NaCl and NP9 in the formation of nanorods are discussed in detail.

2 Experimental

2.1 Materials

 $SnO₂ (>99.5%)$, NaCl (>99.5%), nonyl phenyl ether (NP9) $(>99.0\%)$.

2.2 Synthesis of $SnO₂$ nanorods

 0.5 g of starting SnO₂ powder was mixed with 1.5 g of NaCl and 4 ml of NP9 using an agate mortar and then the mixture was ground for 30 min by hand. The mixed sample was heated in a porcelain crucible that was placed in the middle of an alumina tube with a horizontal tube electric furnace at 800 $^{\circ}$ C for 2.5 h, the heat treatment sample was then cooled gradually to room temperature in air, washed several times with distilled water to remove the NaCl flux, filtered, and then dried in an oven at 70 \degree C for 3 h. The obtained products were collected for characterization by XRD, TEM, HRTEM, SAED, EDS and FTIR.

2.3 Equipment

X-Ray powder diffraction (XRD) patterns were obtained on a Japan Rigaku D_{Max} γ_A X-ray diffractometer with Cu-K α radiation ($\lambda = 0.154178$ nm), employing a scanning rate of 0.02° s⁻¹ in the 2 θ range from 20–80[°]. Transmission electron microscopy (TEM) images and selected area electron diffraction (SAED) patterns were taken with a JEM-200 CX transmission electron microscope, using an accelerating voltage of 200 kV. The samples for these measurements were dispersed in absolute ethanol with an ultrasonic generator, and then the solutions were dropped onto copper grids coated with amorphous carbon films. High-resolution transmission electron microscopy (HRTEM) and energy-dispersive X-ray

spectroscopy (EDS) were carried out on a JEOL-2010 electron microscope, using an accelerating voltage of 200 kV. Infrared spectra were recorded using a Nicolet 170 SX Fourier transform infrared spectroscopy (FTIR) spectrometer from 4000 to 400 cm^{-1} at room temperature. The samples and KBr crystals were ground together, and the mixture was pressed into a disc for recording the spectrum. The 514.5 nm line of the laser was the excitation source, with the capability of supplying 30 mW power.

3 Resuts and discussion

3.1 Characterization of $SnO₂$ nanorods

To characterize the structural features and chemical composition of the nanorods, investigations of XRD, TEM, HRTEM, SAED, EDS, and FTIR were carried out on these nanorods.

Phase identification of as-prepared nanorods was carried out using the XRD pattern shown in Fig. 1. All diffraction peaks can be perfectly indexed with respect to the rutile $SnO₂$ structure, not only in peak position, but also in their relative intensity. XRD patterns showed that the materials were $SnO₂$ with the refined lattice parameters. The average $SnO₂$ lattice constants obtained by refinement of the XRD data for the as-synthesized samples are $a = 4.732 \text{ Å}$ and $c = 3.184 \text{ Å}$, which are consistent with those of bulk $SnO₂$ (JCPDS 21-1250).

The morphology of the nanorods was observed by TEM. A typical TEM image of the as-synthesized nanorods is shown in Fig. 2a. It can be seen that $SnO₂$ crystals display a rod-like shape with a diameter of $20-40$ nm and length of up to 1 μ m. We also find that these nanorods are relatively straight and their surfaces are smooth, and that there are no spherical droplets at their tips. Fig. 2b shows a typical SAED pattern, which confirms that the nanorods are the single crystal rutile $SnO₂$

HRTEM is a powerful method for structural analysis on the atomic scale, and thus might provide further insight into the structure of an individual $SnO₂$ nanorod. Fig. 2c shows an HRTEM image of the as-prepared nanorods. The image shows that $SnO₂$ nanorods are structurally uniform and single crystalline with no defects and dislocations; the interplanar spacing is ca. 0.343 nm, which corresponds to the (110) plane of the rutile crystalline $SnO₂$. The chemical composition analysis by EDS from a local area shows that the nanorods are composed of only Sn and O. Further quantitative analysis reveals that the atomic ratio of Sn : O is 32.75 : 67.25, which is ca. 1 : 2.

Fig. 3 shows the FTIR spectrum of the nanorods. Here, the two characteristic absorption peaks, at 675.3 and 476.8 cm^{-1} , can be attributed to the SnO_2 .²⁸ We find no other peaks in the spectrum. The peak at 675.3 cm^{-1} is assigned to the antisymmetric Sn–O–Sn stretching mode of the surface-bridging oxide formed by condensation of an adjacent hydroxy group; the peak at 476.8 cm⁻¹ is assigned to a symmetric Sn-O–Sn

Fig. 1 X-Ray diffraction pattern of the as-prepared $SnO₂$ nanorods.

Fig. 2 (a) TEM image of $SnO₂$ nanorods; (b) SAED pattern of $SnO₂$ nanorods; (c) HRTEM image of a 20 nm $SnO₂$ nanorod.

stretching mode. Thus the IR spectrum further confirms that the sample is composed of $SnO₂$.

3.2 Growth mechanism of the nanorods

Nanorods synthesized via this annealing process are regular with a relatively straight, rod-like shape. Moreover, there are no spherical liquid droplets at the tips of the nanorods, which are known to be good evidence for the vapor–liquid–solid $(VLS)^{29}$ and solution–liquid–solid $(SLS)^7$ mechanisms. This suggests that the nanorods formed via our method may not grow by the VLS or SLS mechanisms. In order to reveal further the growth mechanism and the effects of NaCl and the surfactant NP9, we performed the following comparative experiments at the same synthesis conditions (the same heating temperature, quantity of each starting material, soaking and grinding times): (a) adding NaCl and NP9, gradually cooling to room temperature (5 $^{\circ}$ C min⁻¹); (b) adding NaCl and NP9, quickly cooling to room temperature (40 °C min⁻¹); (c) adding only NaCl; (d) adding only NP9; (e) no addition of NaCl and NP9. We obtained following experimental results: (1) only experiments (a) and (b) can form nanorods, which are shown in Figs. 4a and b, and the other three experiments cannot form

Fig. 3 FTIR spectrum of the as-prepared $SnO₂$ nanorods.

Fig. 4 (a) TEM images of $SnO₂$ nanorods prepared by adding NaCl and NP9, and gradually cooling to room temperature; (b) TEM images of SnO₂ nanorods prepared by adding NaCl and NP9, and quickly cooling to room temperature; (c) TEM images of $SnO₂$ particles prepared from adding only NaCl; (d) TEM images of $SnO₂$ particles prepared from adding only NP9; (e) TEM images of $SnO₂$ particles prepared by no addition of NaCl and NP9.

nanorods but produce only nanoparticles, which are shown in Figs. 4c–e; (2) the average length and diameter of the nanorods did not show a clear dependence on the cooling rate. These results indicate that the growth mechanism of the nanorods in this route is different with that in the usual 'flux method' in which the product crystals are obtained during the cooling process.30 That is, the nanorods formed via the present method grow not during the cooling process, but during the soaking of the melt. This suggests that the nanorods formed via our method may grow mainly by an Ostwald ripening mechanism, i.e. the dissolving of fine particles and the deposition of components on larger particles.

3.3 Effects of NaCl and surfactant NP9

When Ostwald ripening is the dominant mechanism of nanorod growth, the formation of the nanorods must be affected by the character of the starting material, such as the particle size and/or chemical activity, because the dissolution rate of the material depends on such characters. The viscosity of the flux during heating of the system also affects the for the riax during neuting of the system also arrects the formation of nanorods, 31 and these points can be confirmed by the above-mentioned experiments. In these experiments, only (a) and (b) can form nanorods (Figs. 4a, b), whereas experiments (c), (d) and (e) all produce nanoparticles (Figs. 4c–e). These results may give an explanation of the effects of NaCl and NP9 in the formation of nanorods. Since adding NaCl can significantly decrease the viscosity of the melt, 31 this makes the mobility of components in the flux easier. So, adding NaCl can provide a favorable environment for the growth of nanorods. The surfactant NP9 may play the following role: during the grinding process of the precursor $SnO₂$ particles, adding the NP9 is favorable to the formation of fine particles because the NP9 may form a 'shell' surrounding the particles to prevent them from aggregating to larger particles, thus decreasing the size of the precursor $SnO₂$ particles. Fig. 5a shows the TEM image of ground precursor $SnO₂$ particles. The average diameter of as-ground $SnO₂$ particles is *ca*. 100 nm, but most particles with a diameter of less than 100 nm can also be found. These particles display an irregular shape. Fig. 5b shows TEM images of $SnO₂$ particles before grinding. It can be seen that the average diameter of the unground $SnO₂$ particles is *ca*. 150 nm. Some particles with a diameter of more than 300 nm can be found. The unground $SnO₂$ particles also have an irregular form. Consequently, ground precursor powders have highlyspecific surfaces and are metastable in the annealing process, which makes it easy to further fabricate nanorods. This is also

Fig. 5 TEM images of (a) ground precursor $SnO₂$ particles; (b) $SnO₂$ particles before grinding.

favorable for the formation of nanorods. That is why the nanorods can grow only in (a) and (b) experiments, but not in the cases (c) , (d) and (e) .

4 Conclusions

A simple annealing process has been successfully developed to synthesize SnO₂ nanorods. These nanorods have a diameter of $20-40$ nm and length of up to 1 μ m. The structural nature and chemical composition of the nanorods were analyzed by XRD, TEM, HRTEM, EDS and FTIR. In comparison to the mechanism of some other methods for the fabrication of onedimensional nano-scale materials, the formation mechanism of nanorods by our method is different. We propose that the nanorods prepared using this method might fabricate mainly by an Ostwald ripening mechanism. This method is simple and controllable, and thus, the product SnO₂ nanorods can be of high purity.

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